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Isotope tuning of the superconducting dome of strontium titanate

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Doped strontium titanate SrTiO_3 (STO) is one of the most dilute superconductors known today. The fact that superconductivity occurs at very low carrier concentrations is one of the two reasons that the pairing mechanism is not yet understood, the other being the role played by the proximity to a ferroelectric instability. In undoped STO, ferroelectric order can in fact be stabilized by substituting ^{16}O with its heavier isotope ^{18}O . Here, we explore the superconducting properties of doped and isotope-substituted $\text{SrTi}({}^{18}\text{O}_y{}^{16}\text{O}_{1-y})_{3-\delta}$ for $0 \leq y \leq 0.81$ and carrier concentrations between 6×10^{17} and $2 \times 10^{20} \text{ cm}^{-3}$ ($\delta < 0.02$). We show that the superconducting T_c increases when the ^{18}O concentration is increased. For carrier concentrations around $5 \times 10^{19} \text{ cm}^{-3}$ this T_c increase amounts to almost a factor 3, with T_c as high as 580 mK for $y = 0.74$. When approaching $\text{SrTi}^{18}\text{O}_3$ the maximum T_c occurs at much smaller carrier densities than for pure $\text{SrTi}^{16}\text{O}_3$. Our observations agree qualitatively with a scenario where superconducting pairing is mediated by fluctuations of the ferroelectric soft mode.

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I. INTRODUCTION

Liquid helium does not become solid at ambient pressure, even for temperatures close to absolute zero. Solidification is inhibited by the zero-point fluctuations (ZPFs) of the He atoms due to their small nuclear mass. A similar effect comes into play in the cubic perovskite SrTiO_3 (STO), which would be ferroelectric in the absence of ZPF [1]. Experimentally, the material is instead observed to be a quantum paraelectric insulator with a large dielectric constant ($\epsilon \sim 10^4$) as the ferroelectric ground state is suppressed by the ZPF of the oxygen atoms [2]. This is a quantum effect and the ZPF amplitude is controlled by $\hbar^2/m$ where m is the ionic mass. Tuning $\hbar$ is impossible, but substituting more than 33 at. % of the natural ^{16}O isotope by the heavier ^{18}O reduces the ZPF sufficiently to stabilize the ferroelectric phase [3,4]. Alternative methods for tuning ferroelectric order in STO are Ca substitution [5] and strain [6]. The advantage of the isotope substitution route is that the stoichiometry remains unaffected and modification of the electronic properties is fully obtained through quantum tuning of the ZPF amplitude. Doping a small density of electrons in STO induces superconductivity with a superconducting transition temperature T_c ranging between 50 mK (carrier density 10^{17} cm^{-3}) and 400 mK (10^{20} cm^{-3}) [7–10]. Isotope substitution is also a key test of the BCS theory of superconductivity. In STO the sign of the isotope effect on T_c is opposite to the BCS prediction and the observed isotope effect is an order of magnitude stronger than the BCS prediction [11,12], a state of affairs that has been anticipated by Edge *et al.* based on a model where soft ferroelectric fluctuations

provide the pairing interaction for superconductivity carriers [1]. Further indications that T_c is enhanced in proximity to ferroelectricity comes from strain experiments [13,14]. Due to symmetry selection rules, the coupling to the ferroelectric modes may be too weak to be compatible with experimental T_c values [15]. However, coupling to pairs of these modes is not ruled out by symmetry considerations and in STO this coupling is in fact quite strong [16,17].

Here, we provide a broad coverage of the two-dimensional parameter space of ^{18}O isotope substitution (up to 81%) and carrier density (6×10^{17} – $2 \times 10^{21} \text{ cm}^{-3}$) and show how the superconducting dome of STO changes as the system is tuned via ^{18}O substitution across its quantum critical point (QCP) into the ferroelectric phase (33 at. % ^{18}O for undoped STO). Our results challenge the concept of pairing mediated by phonons in the sense of bosonic particles, and require a generalization of the boson concept in the presence of strong anharmonicity.

II. FERROELECTRIC ORDER

We fabricated isotope-substituted $\text{SrTi}({}^{18}\text{O}_y{}^{16}\text{O}_{1-y})_3$ samples by heating commercial STO crystals in an $^{18}\text{O}_2$ atmosphere (see Appendix A for more details). As-substituted samples are insulating and their dielectric constant as a function of temperature is shown in Fig. 1(a). Above a certain ^{18}O level, a peak appears in the dielectric constant marking the Curie temperature T_{Curie} . Figure 1(b) shows the paraferroelectric phase diagram, i.e., T_{Curie} as a function of y , and confirms previous works that found a ferroelectric order above $y = 0.33$ [3,18].

Below 110 K SrTiO_3 is tetragonal, and has four infrared-active modes. Three modes at 190, 420 and 500 cm^{-1} (TO2, TO3, and TO4, respectively) show less than 6% redshift for complete ^{18}O substitution as a result of the heavier mass. The soft ferroelectric mode TO1 at 8 cm^{-1} is by far the

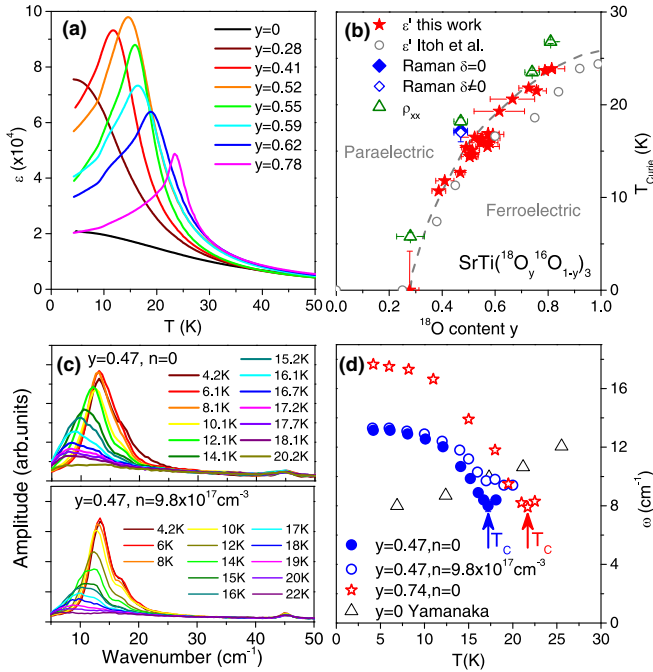


FIG. 1. Ferroelectricity in $\text{SrTi}^{(18}\text{O}_y^{16}\text{O}_{1-y})_3$. (a) Dielectric constant as a function of temperature. (b) Para-ferroelectric phase diagram including previous data by Itoh *et al.* [18]. (c) Low-temperature Raman spectra for $y = 0.47$ showing the hardening of the ferroelectric soft mode below the T_{Curie} . (d) Temperature dependence of the Raman-detected soft mode for samples with $y = 0.47$ (undoped and doped) and 0.74 (undoped) compared to hyper-Raman measurements for undoped $y = 0$ [19].

most strongly affected due to the strong anharmonicity of this mode. Raman spectroscopy of the TO1 mode provides additional experimental evidence for ferroelectricity in the isotope-substituted samples. Figure 1(c) shows Raman spectra for $y = 0.47$ displaying the soft mode that becomes Raman active below T_{Curie} and then shows a hardening with decreasing temperature [see Fig. 1(d)] that is not present for $y = 0$ [19]. The fact that the ferroelectric soft mode becomes Raman active signals the breaking of inversion symmetry at the Ti sites, which is consistent with ferroelectric order.

In a second step isotope-substituted samples with $y = 0, 0.28 \pm 0.04, 0.52 \pm 0.02, 0.74 \pm 0.04$, and 0.81 ± 0.05 are annealed in vacuum to remove part of the oxygen atoms to introduce n -type carriers and make the system metallic. This diffusive process removes ^{16}O and ^{18}O atoms to a different degree due to a (small but finite) difference in the diffusion constant. The effect on the $^{18}\text{O}/^{16}\text{O}$ ratio is negligible because only 10^{-5} – 10^{-2} oxygen atoms/u.c. are removed to make the samples metallic, as compared to $y = 0.3$ needed for ferroelectricity. Our metallic $\text{SrTi}^{(18}\text{O}_y^{16}\text{O}_{1-y})_{3-\delta}$ samples display quantum oscillations of the same frequency as previous studies of $\text{SrTi}^{16}\text{O}_{3-\delta}$ with the same carrier density [10,23] (see Appendix C). This implies the existence of a well-connected Fermi sea excluding a collection of metallic puddles and shows that the shape of the Fermi surface is not altered upon ^{18}O substitution. Besides the quantum oscillations, other normal-state properties such as low-temperature mobility do not change upon isotope substitution (see Appendix D).

For the soft ferroelectric mode we observe the same temperature dependence of the Raman spectra in the electron-doped material as in the undoped sample [Figs. 1(c) and 1(d)]. This confirms that in these materials ferroelectric inversion symmetry breaking and free carriers can coexist [21,24,25]. Since the doped materials are not true ferroelectrics due to the presence of free charge carriers, we will use the label “ferroelectriclike” in those cases.

Below 100 K, the resistivity of n -doped STO shows a quadratic temperature dependence of the form $\rho_{xx} = \rho_0 + AT^2$. The residual resistivity ρ_0 describes the elastic scattering by disorder and the doping-dependent prefactor A the inelastic scattering.

Figures 2(a)–2(d) show the normal-state resistivity as a function of T and T^2 of selected substituted samples. At low temperatures, ρ_{xx} of samples with carrier density below a certain critical threshold n^* deviates from a T^2 behavior and shows a doping-dependent anomaly that is absent in unsubstituted STO. As can be seen in Fig. 1(b), the resistivity anomaly occurs around the Curie temperature determined by permittivity and Raman spectroscopy measurements indicating a common cause of resistivity anomaly and Raman activation of the soft ferroelectric (FE) mode. In Fig. 2(e) the temperature of this anomaly (see Appendix E) is displayed as a function of carrier density. With increasing n , the anomaly shifts to lower temperatures until it disappears at n^* at which the resistivity recovers a T^2 behavior and the prefactor A is similar for unsubstituted and substituted samples [see Fig. 2(d)]. Figure 2(f) shows n^* as a function of y . For ^{18}O isotope-substituted $\text{La}_{1-x}\text{Sr}_x\text{TiO}_3$ such an anomaly has not been reported [12], but for $\text{Sr}_{1-x}\text{Ca}_x\text{TiO}_{3-\delta}$ similar features in the temperature dependence of resistivity, thermal expansion, and sound velocity, as well as the activation of the TO1 mode in the Raman spectrum, indicate that a ferroelectriclike phase persists in the metallic state [21,24] and that this phase disappears once a critical threshold of around eight electrons/dipole is reached. The dependence on isotope concentration and carrier density in Fig. 2(f) indicates that this symmetry breaking is still present for relatively high carrier concentration $n = 2 \times 10^{18} \text{ cm}^{-3}$ and low isotope concentration $y = 0.28$. The length scale of this ferroelectriclike order cannot be determined from the present experiments.

III. SUPERCONDUCTIVITY

Figures 3(a)–3(d) show the resistive superconducting transitions at selected carrier densities n for the different isotope levels. There is a clear enhancement of the superconducting transition temperature upon isotope substitution. To rule out any other reason for T_c enhancement other than isotope substitution, twin samples have been annealed in $^{16}\text{O}_2$ atmosphere under the same conditions as in the ^{18}O substitution process. No enhancement of T_c was detected in these samples and their superconducting properties agree with previous works [23]. Figure 3(e) shows the bulk superconducting transition detected in ac susceptibility for a sample with $y = 0.81$. For strontium titanate the T_c measured by bulk probes (susceptibility, specific heat, and thermal conductivity) is always lower than the T_c from resistivity measurements [21,26]. The most plausible reason is heterogeneity of the resistive T_c

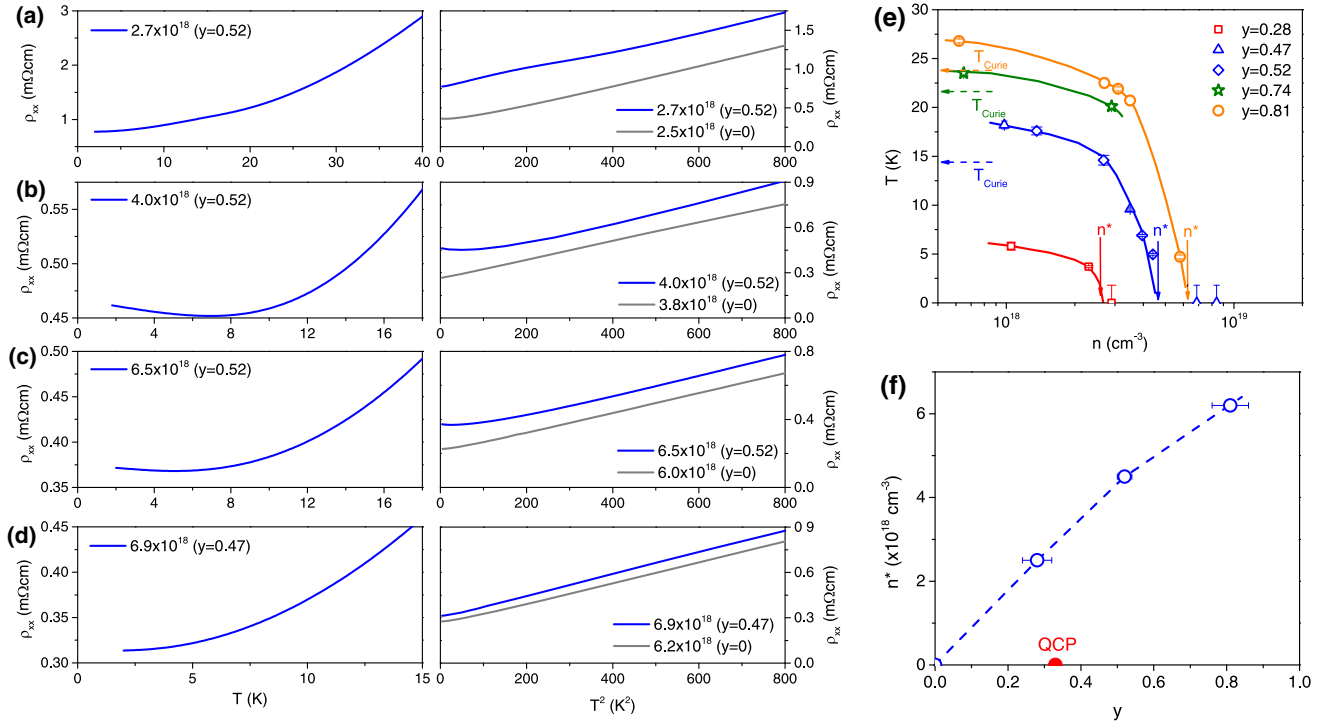


FIG. 2. Temperature dependence of resistivity. (a)–(d) Resistivity ρ_{xx} as a function of T and T^2 for samples with $y \approx 0.5$ and different n . For comparison, ρ_{xx} of samples with $y = 0$ and similar n is shown as well as a function of T^2 . (e) Temperature of the anomaly in resistivity associated with the polar order. Horizontal arrows mark T_{Curie} measured on the insulating samples with the same ¹⁸O content. (f) Critical carrier density n^* as a function of y .

within the samples, possibly associated to the formation of tetragonal domains below 110 K. The superconducting critical temperatures T_c , defined as the midpoint of the resistive transitions, are plotted in the superconducting phase diagram in Fig. 4(a). ¹⁸O substitution has a significant influence on both the maximum value of T_c as well as the shape of the entire dome. Upon ¹⁸O substitution, the maximum T_c is continuously enhanced with increasing substitution level. Whereas T_c on the overdoped side of the dome remains mostly unchanged, there is a large increase on the underdoped side of the dome ($5 \times 10^{18} \lesssim n \lesssim 10^{20} \text{ cm}^{-3}$). This leads not only to a slight shift of the dome maximum to lower doping, but a very strong broadening of the entire dome. This increase of T_c limited to the underdoped side of the dome is clearly visible in the color plot of T_c in Fig. 4(b). The increase already appears at $y = 0.28 \pm 0.04$, i.e., around the quantum critical point confirming earlier results on $y = 0.35$ samples [11]. Below $5 \times 10^{18} \text{ cm}^{-3}$ the effect of isotope substitution diminishes.

For a given carrier concentration, isotope substitution reduces the energy gap of the polar TO1 mode frequency until the gap vanishes at a critical value. Simultaneously, T_c increases on the underdoped side of the dome. Applying pressure has the opposite effect to ¹⁸O substitution as it forces the polar TO1 mode to harden and would therefore correspond to a negative isotope effect in Fig. 4(b). Pressure experiments have shown that T_c decreases linearly as a function of pressure [27] and that superconductivity is eventually destroyed at 5.5 kbar for $\text{SrTi}_{1.98}\text{Nb}_{0.02}\text{O}_3$ ($n = 3.4 \times 10^{19} \text{ cm}^{-3}$) [22]. Figure 4(c) compares the data obtained for ¹⁸O- and Ca-substituted STO together with the pressure studies as it plots

normalized T_c as a function of the inverse dielectric constant $1/\epsilon_0$ taken at low temperature from the undoped systems.

IV. CONCLUSIONS

An important question is how the isotope substitution affects the superconducting properties, and what the implications of our experiments are for the mechanism of superconductivity in SrTiO_3 . The trend of increasing T_c upon isotope substitution and decreasing T_c upon applying pressure corresponds qualitatively to the behavior expected when pairing is mediated by the soft ferroelectric mode [1]. Although the coupling to the LO1 mode at 22 meV is very small ($\lambda = 0.003$) [28,29], the main features of the phase diagram have been reproduced in a hybrid model of coupling to plasmons and the LO1 mode [15,22,30,31]. In this approach the electron-electron interaction is treated in the random phase approximation. Since this approximation has been shown to strongly overestimate the stability of the superconducting state [32], further analysis of the hybrid phonon-plasmon model is required taking into account exchange and correlation terms. The key experimental observations of this work is the isotope-induced shift of the superconducting dome to lower carrier concentrations and the quantum critical line in the doping-isotope plane confirming the theoretical predictions by Edge *et al.* [1]. Increasing the ¹⁸O isotope content from 33% to 81% increases the critical carrier concentration from $n = 0$ to $n = 10^{19} \text{ cm}^{-3}$. The T_c enhancement in the underdoped region is caused by the confluence of two trends, namely the fact that T_c tends to 0 in the limit of zero carrier

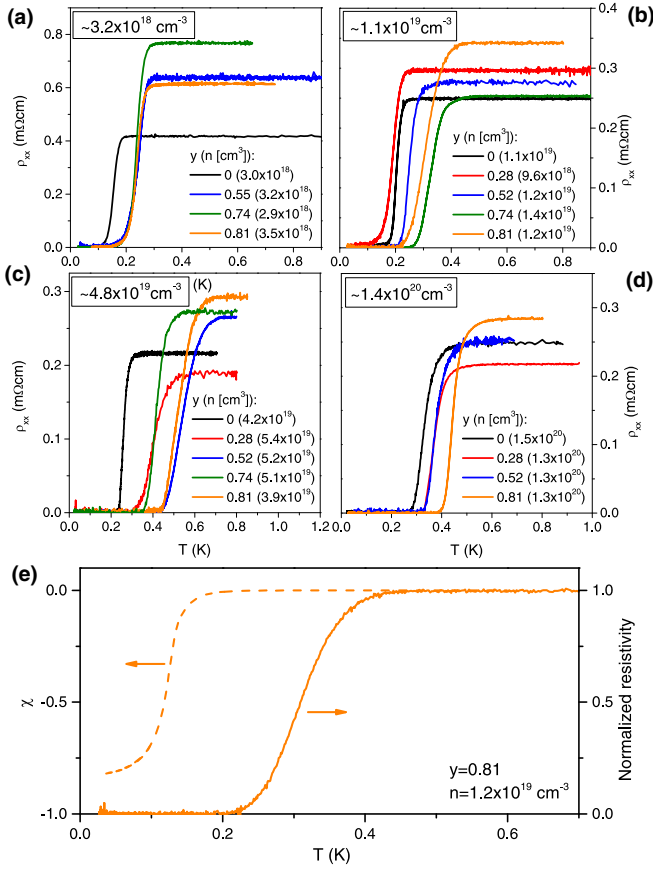


FIG. 3. Superconducting transitions. Superconducting transitions seen in resistivity measurements at selected carrier densities of (a) 3.2×10^{18} , (b) 1.1×10^{19} , (c) 4.8×10^{19} , and (d) $1.4 \times 10^{20} \text{ cm}^{-3}$ (see Appendix B for all resistive transitions). (e) Superconducting transition seen by ac magnetic susceptibility and resistivity for $y = 0.81$ and $n = 1.2 \times 10^{19} \text{ cm}^{-3}$.

density and the enhancement of T_c at the ferroelectric quantum critical point. Ruhman and Lee [15] pointed out that the coupling to the TO1 phonon [1] is very small and vanishes in the limit of zero doping. This argument does not apply to pairs of TO1 phonons [16,33], which are expected to couple quite strongly to the electrons and may constitute the main pairing mechanism at low doping [17].

The data sets generated and analyzed during the current study are available in Ref. [34] and will be preserved for 10 years.

ACKNOWLEDGMENT

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APPENDIX A: FURTHER INFORMATION ON EXPERIMENTAL METHODS

In this study we have used commercial $\text{SrTi}^{16}\text{O}_3$ substrates obtained from different suppliers (MTI, Crystech GmbH, and Crystal GmbH). The samples have thicknesses of 0.1 or 0.25 mm and a (100) or (110) crystal orientation. The

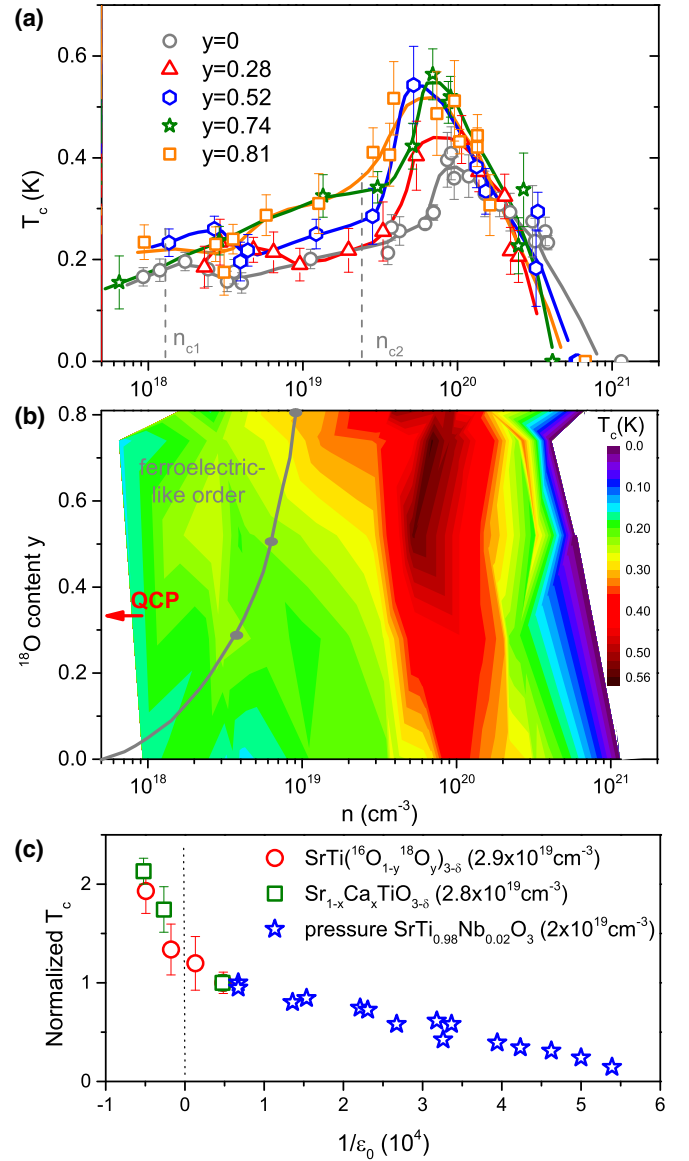


FIG. 4. Superconducting phase diagrams. (a) Superconducting T_c as a function of carrier density n . Dashed lines mark critical densities n_{c1} and n_{c2} at which the second and third band are filled, respectively [10,20]. (b) Color plot showing the ferroelectric and the superconducting order as a function of n and y . The gray line corresponds to the critical threshold n^* [see Fig. 2(e)]. (c) Normalized T_c vs $1/\epsilon_0$ taken at low temperature on the insulating samples for oxygen-depleted $\text{SrTi}^{18}\text{O}_{1-y}\text{O}_{3-\delta}$, $\text{Sr}_{1-x}\text{Ca}_x\text{TiO}_{3-\delta}$ (Ref. [21] and, in the interest of completeness, an extra measurement for $x = 0.0022$), and $\text{SrTi}_{0.98}\text{Nb}_{0.02}\text{O}_3$ under pressure [22]. Negative $1/\epsilon_0$ signifies that insulating samples ($\delta = 0$) with the same x or y are ferroelectric. T_c was normalized to the value obtained for $y = 0$, $x = 0$, or zero pressure.

substitution process has been done in standard quartz tubes that were filled with ^{18}O gas to pressures of 0.4–0.7 bars and then sealed with a flame torch. The sealed tubes have been heated using different furnaces at temperatures between 700 and 1100 °C for at least 20–30 days. The annealing time t has been chosen in a way that the diffusion path calculated using $\sqrt{2Dt}$, with D the self-diffusion coefficient of

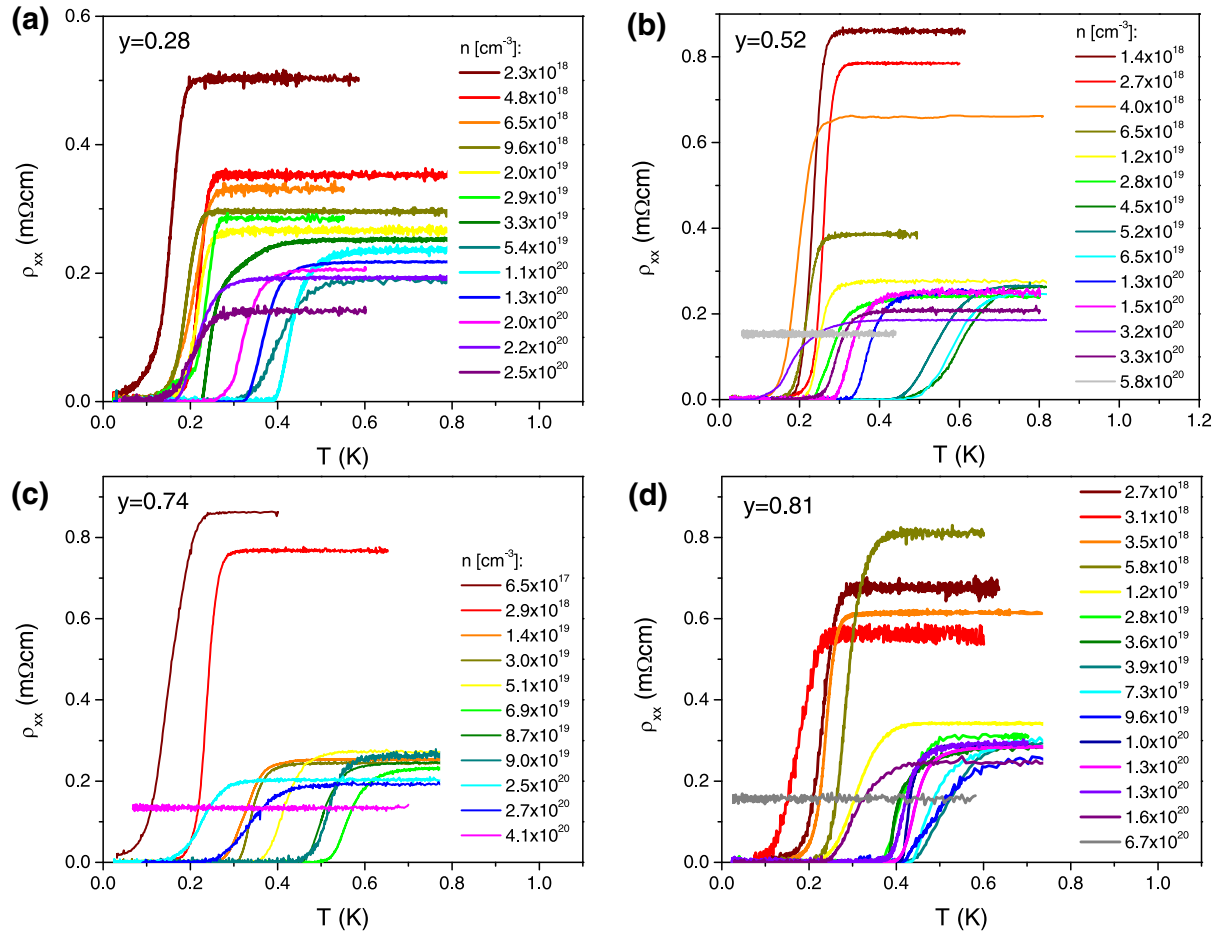


FIG. 5. Resistive superconducting transitions. Resistivity ρ_{xx} as a function of temperature for all measured isotope-substituted $\text{SrTi}^{(18}\text{O}_{1-y})_{3-\delta}$ samples.

oxygen in strontium titanate [35,36], amounts to at least half of the sample thickness. The final ^{18}O concentration in the samples was controlled by the amount of ^{18}O gas in the tube and the number of times the substitution process was repeated.

As in previous studies [3], the amount of ^{18}O in the substituted substrates has been determined by the mass enhancement due to the heavier ^{18}O isotope. The mass of the sample has been measured before and after each substitution process and the substitution rate has been calculated from the mass increase using

$$y_f = \frac{m_f[y_i M_{18} + (1 - y_i)M_{16}] - m_i M_{16}}{m_i(M_{18} - M_{16})}, \quad (\text{A1})$$

with y_i and y_f the initial and final ^{18}O content before and after the substitution process, m_i and m_f the mass of the sample before and after the substitution process, and M_{16} and M_{18} are the molar masses of $\text{SrTi}^{16}\text{O}_3$ and $\text{SrTi}^{18}\text{O}_3$, respectively. The balance used to measure the weight increase had a precision of 0.01 mg. The errors of the ^{18}O concentration indicated in the main text stem from the mass variation found in weighing each sample subsequently multiple times.

For the dielectric measurements gold electrodes have been evaporated on both sides of the insulating samples and con-

tacted with silver paste. The capacitance of the samples was measured from 300 to 4.2 K in a home-built cryostat with a high-precision LCR meter (Agilent E4980A) using typically an excitation of 50 mV and 1 kHz.

For the electrical measurements (four-point linear resistivity and Hall effect), gold contact pads were evaporated prior to doping by vacuum annealing. Electrical measurements in the temperature range of 1.8–300 K have been done in a Quantum Design physical property measurement system (PPMS). The superconductivity of the samples was studied in a dilution refrigerator with a base temperature of 25 mK. Magnetic ac susceptibility measurements were performed in the dilution fridge with a home-built setup using an excitation and a pickup coil. A lock-in amplifier with a low-noise preamplifier was used to both create the ac excitation current and to read the voltage induced in the pickup coil. The amplitude of the ac excitation fields B was of the order of 1–5 μT and had a frequency f in the range of a few kHz. The susceptibility χ has been calculated from the measured voltage ΔV drop using $\Delta V = 2\pi^2 N f \eta \chi B r^2$ with N the number of turns, r the radius of the pickup coil, and the filling factor η is given by the ratio of the sample volume to the inner volume of the pickup coil [37]. It was checked that the measured voltage drop was proportional to both B and f over a wide range of excitation parameters.

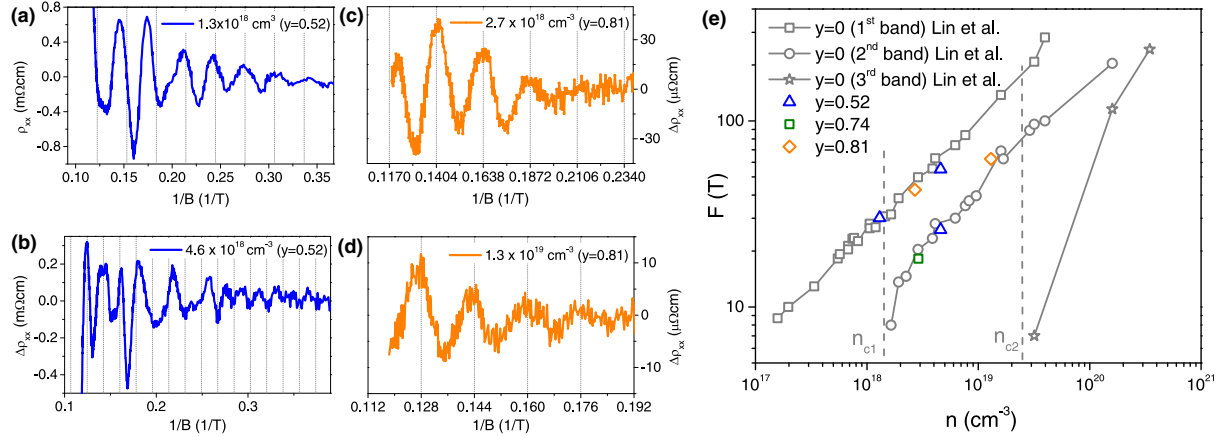


FIG. 6. Quantum oscillations measured in $\text{SrTi}({}^{18}\text{O}_y{}^{16}\text{O}_{1-y})_{3-\delta}$ samples with (a), (b) $y = 0.52$ and (c), (d) $y = 0.81$ samples and different carrier densities at a temperature of 30 mK. (e) Oscillation frequencies F as a function of carrier density n compared to values measured by Lin *et al.* [9,23] on unsubstituted $\text{SrTi } {}^{16}\text{O}_{3-\delta}$ ($y = 0$).

APPENDIX B: SUPERCONDUCTING RESISTIVE TRANSITIONS

Figure 5 shows the superconducting transitions of all samples displayed in the phase diagram in the main text. The values of T_c shown in the phase diagram correspond to the midpoint of the transition, i.e., the temperature where resistivity is at 50% of its normal-state value. The error bars of T_c have been defined as half of the temperature difference between the temperatures corresponding to 10% and 90% of the normal-state resistivity.

APPENDIX C: QUANTUM OSCILLATIONS

Quantum oscillations in selected samples have been studied at dilution fridge temperatures in magnetoresistance measurements up to 8.5 T. Figures 6(a)–6(d) show the oscillating part $\Delta\rho_{xx}$ of the magnetoresistance, obtained after subtraction of a smooth background, as a function of inverse magnetic field $1/B$. Figure 6(e) plots the detected oscillation periods F as a function of carrier density in comparison with frequencies measured by Lin *et al.* [9,23] on $\text{SrTi } {}^{16}\text{O}_{3-\delta}$. These previous studies of the Fermi surface of doped STO identified two critical doping levels at $n_{c1} \approx 1.5 \times 10^{18}$ and $n_{c2} \approx 3 \times 10^{19} \text{ cm}^{-3}$ corresponding to the filling of a second and third band, respectively. Furthermore, they showed that the displayed frequencies match the carrier densities obtained from Hall effect measurements. As shown in Fig. 6, the oscillation periods measured in $\text{SrTi}({}^{18}\text{O}_y{}^{16}\text{O}_{1-y})_{3-\delta}$ samples agree well with those observed for unsubstituted STO, implying that our electron-doped isotope-substituted samples thus have a single well-connected Fermi surface with the same shape as observed in unsubstituted $\text{SrTi } {}^{16}\text{O}_{3-\delta}$.

APPENDIX D: NORMAL-STATE ELECTRIC TRANSPORT

The low-temperature mobility $\mu = (en\rho_{xx})^{-1}$ of all investigated samples is shown in Fig. 7(a). In unsubstituted n -doped STO, the mobility is proportional to $\propto n^{-5/6}$ [38]. In ${}^{18}\text{O}$ -substituted samples, μ shows a deviation from this power-law

behavior starting below the critical carrier threshold n^* identified in the main text, i.e., the mobility of the electrons is reduced in the presence of the polar order. Figure 7(b) plots the mean free path $l = \hbar\mu k_F/e$ with the Fermi wave vector $k_F = (3\pi^2 n)^{1/3}$ obtained by assuming a spherical Fermi surface. Both of the observed features, i.e., a deviation of μ from

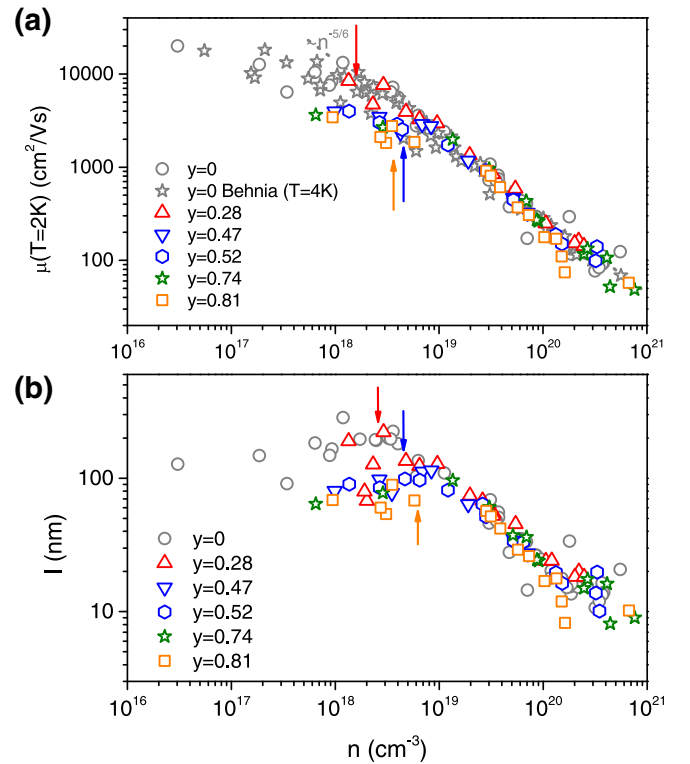


FIG. 7. (a) Low-temperature mobility $\mu(2 \text{ K})$ as a function of carrier density for all samples. The data for $y = 0$ are compared to data from Ref. [38] (measured at 4 K). (b) Mean free path l as a function of carrier density. Arrows in both panels mark the critical carrier density n^* shown in Fig. 2 of the main text.

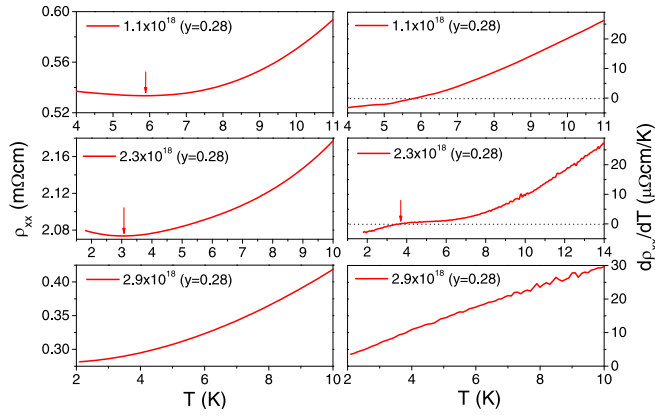


FIG. 8. Resistivity ρ_{xx} (left-hand side) as well as its derivative $d\rho_{xx}/dT$ (right-hand side) of samples with an ^{18}O content of $y = 0.28$ at different carrier densities n .

a power-law dependence below n^* and a broadened maximum of l around n^* , show that the elastic scattering of the electrons is affected by the polar order that persists below n^* . A similar behavior of μ and l has been observed in $\text{Sr}_{1-x}\text{Ca}_x\text{TiO}_{3-\delta}$ samples [21,24].

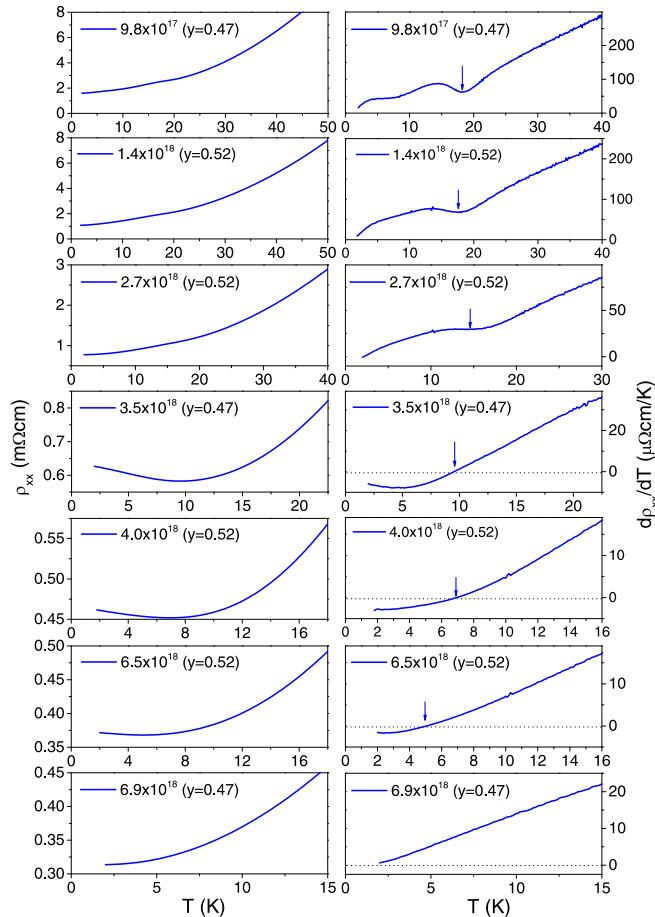


FIG. 9. Resistivity ρ_{xx} (left-hand side) as well as its derivative $d\rho_{xx}/dT$ (right-hand side) of samples with an ^{18}O content of $y = 0.47$ or $y = 0.52$ at different carrier densities n .

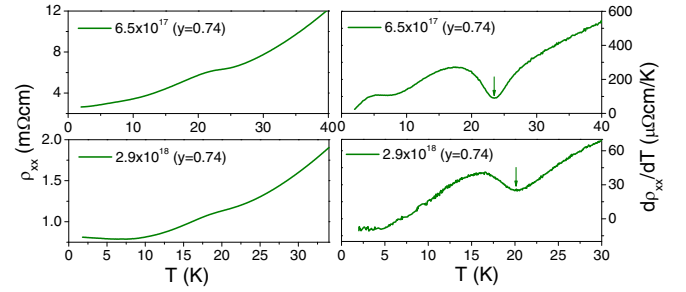


FIG. 10. Resistivity ρ_{xx} (left-hand side) as well as its derivative $d\rho_{xx}/dT$ (right-hand side) of samples with an ^{18}O content of $y = 0.74$ at different carrier densities n .

APPENDIX E: TEMPERATURE DEPENDENCE OF RESISTIVITY

Figures 8–11 show the resistivity ρ_{xx} as well as its derivative $d\rho_{xx}/dT$ for ^{18}O contents $y = 0.28, 0.52, 0.74$, and 0.81 , respectively, as a function of temperature. Arrows mark the temperature of the anomaly associated with the Curie temperature ferroelectriclike transition at which ρ_{xx} deviates from a T^2 behavior, which is plotted in Fig. 2 of the main text. As the shape of the anomaly depends strongly on the doping level and changes from a hump or kink at lower doping [see, for example, Fig. 2(a) of the main text] to a minimum followed by an upturn of the resistivity [see, for example, Fig. 2(b) of the main text], we used the following criteria to define the temperature of the anomaly: for samples that show a clear minimum followed by an upturn, we have taken the temperature of the minimum of ρ_{xx} (i.e., the temperature for which $d\rho_{xx}/dT = 0$). For samples that show a hump or kink in the resistivity

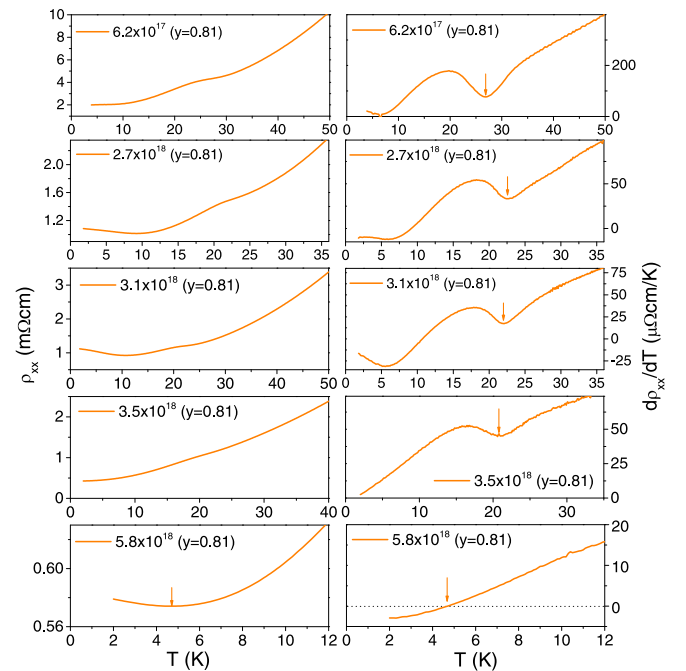


FIG. 11. Resistivity ρ_{xx} (left-hand side) as well as its derivative $d\rho_{xx}/dT$ (right-hand side) of samples with an ^{18}O content of $y = 0.81$ at different carrier densities n .

without a minimum, we have taken the temperature of this hump/kink in $d\rho_{xx}/dT$ for this as the hump/kink is more visible in the derivative. Even though other ways of defin-

ing the temperature of the anomaly are possible, it does not change the fact that the anomaly disappears at a critical doping threshold n^* .

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